

Methane Coupling Reaction on Oxide Solid Solution Catalysts<sup>1</sup>

C. B. ALCOCK,\* J. J. CARBERRY,† R. DOSHI,\* AND N. GUNASEKARAN†

*\*Center for Sensor Materials, Department of Electrical Engineering, and †Laboratory for Catalysis, Department of Chemical Engineering, University of Notre Dame, Notre Dame, Indiana 46556*

Received January 25, 1993; revised May 10, 1993

The methane coupling reaction to form C<sub>2</sub> hydrocarbons has been studied over perovskite-type oxides La<sub>1-x</sub>Sr<sub>x</sub>B<sub>1-y</sub>M<sub>y</sub>O<sub>3-δ</sub> (B = Co, Cr, Fe, Mn, Y; M = Nb, Ti), K<sub>2</sub>NiF<sub>4</sub>-type superconducting oxides A<sub>2-x</sub>B<sub>x</sub>CuO<sub>4-δ</sub> (A = La, Nd; B = Sr, Ce), and the solid solution La<sub>0.9</sub>Sr<sub>0.1</sub>O<sub>1.45</sub>. The reaction was performed in a fixed bed flow reactor with a feed containing 30% methane, 5% oxygen, and balance helium at 795°C. The results indicated that the superconducting oxides and the perovskites with high p-type electronic conductivity were not active for the methane conversion to C<sub>2</sub> hydrocarbons. The compounds with n-type conductivity, electrolytes, and insulators were all found to proceed with selective oxidation to C<sub>2</sub> compounds. A combination of an insulator or electrolyte associated with increased oxygen vacancy concentration appears to increase the C<sub>2</sub> yield. Electrical conductivity, oxygen diffusion coefficient, and oxygen vacancy concentrations of the materials were considered to explain the observed variations in the methane coupling reactivity. © 1993

Academic Press, Inc.

## INTRODUCTION

Perovskite oxides of the formula ABO<sub>3</sub> (A = La, Sr; B = Cr) and other oxides of the general formula A<sub>2</sub>BO<sub>4</sub> (A = La, Sr; B = Cu) have been shown to be catalytically active for CO oxidation (1-3). All the oxides of the type La<sub>1-x</sub>Sr<sub>x</sub>BO<sub>3-δ</sub> (B = Cr, Co, Fe, Mn) are good electronic conductors (4-7). The mode of electrical conduction of these oxides and of the La<sub>2-x</sub>Sr<sub>x</sub>CuO<sub>4-δ</sub> system is p-type. It has recently been shown that the n-type superconducting oxide system, Nd<sub>2-x</sub>Ce<sub>x</sub>CuO<sub>4-δ</sub> (8), is less active for CO oxidation than the p-type conductors, and the perovskite electrolyte, La<sub>0.8</sub>Sr<sub>0.2</sub>YO<sub>3-δ</sub> (8), is catalytically inactive in this regard. Other studies on the oxidation of hydrocarbons (9, 10) indicate that the perovskite oxide catalysts are comparable in their activity to Pt/alumina catalysts. While some metal oxides such as NiO and MnO<sub>2</sub> are known to be active catalysts for hydrocarbon ox-

dation, it has been shown that combinations of metal oxides can be better catalysts than the component oxides (11). Catalytic oxidation is an important property for the oxidation of undesirable components in exhaust gases, whereas oxidative coupling of methane to produce higher hydrocarbons is important in the use of the abundant quantities of methane available in natural gas, and so both aspects of catalysis are important.

The oxidative coupling of methane has been observed for several oxide catalyst systems (12, 13) and efforts have been made to increase the activity by "promoting" the oxides with other metals such as Li on MgO. Many of these efforts have been made on materials of which the structure has not been well characterized from the crystallographic point of view.

In this study we measured the catalytic properties of well characterized bulk solids whose structure, stability, and electrical properties are well known. Perovskite oxides of the general formula La<sub>1-x</sub>Sr<sub>x</sub>B<sub>1-y</sub>O<sub>3-δ</sub> (B = Co, Cr, Fe, Mn, Y; M = Nb, Ti), oxides in the superconducting systems

<sup>1</sup> Presented at the Symposium on Natural Gas Upgrading, ACS Meeting, April, 1992.

$A_{2-x}B_x\text{CuO}_{4-\delta}$  ( $A = \text{La, Nd}$ ;  $B = \text{Sr, Ce}$ ), and oxides in the La–Sr–Y–O system were examined for oxidative coupling of methane and the results compared with the known defect and electronic properties of these oxides. The compounds included in this study encompass a wide range of electrical properties from superconducting oxides ( $\text{La}_{2-x}\text{Sr}_x\text{CuO}_{4-\delta}$ ,  $\text{Nd}_{2-x}\text{Ce}_x\text{CuO}_{4-\delta}$ ), highly semiconducting p-type oxides ( $\text{La}_{0.8}\text{Sr}_{0.2}\text{BO}_{3-\delta}$ ;  $B = \text{Co, Cr, Mn}$ ), an n-type semiconductor ( $\text{LaFe}_{0.8}\text{Nb}_{0.2}\text{O}_{3-\delta}$ ), and a purely ionic conductor ( $\text{La}_{0.8}\text{Sr}_{0.2}\text{YO}_{2.9}$ ), to insulators ( $\text{La}_2\text{O}_3$ ,  $\text{LaYO}_3$ ,  $\text{Y}_2\text{O}_3$ , and  $\text{SrO}$ ). This study is not an optimized performance evaluation of catalysts but an attempt to evaluate the characteristics of catalytic activity in relation to the well known bulk properties of the catalysts.

#### EXPERIMENTAL

##### (i) Preparation of Oxides by Solid-State Processing

All the oxides except  $\text{LaYO}_3$ -bixbyite and  $\text{La}_{0.9}\text{Sr}_{0.1}\text{O}_{1.45}$  were prepared by the classical solid-state processing method. The component oxides were mixed in the proper proportions, pressed into pellets of 1 cm diameter under a load of 20,000 lbs and sintered at 1100°C. The pellets were then crushed and ground, and the process was repeated several times until a single-phase material was obtained.

$\text{LaYO}_3$  and  $\text{La}_{0.8}\text{Sr}_{0.2}\text{YO}_{3-\delta}$  were prepared by this method at a sintering temperature of 1050°C to form a perovskite phase.

##### (ii) Preparation of Oxides by the Pechini Method

$\text{LaYO}_3$  with the bixbyite structure and  $\text{La}_{0.9}\text{Sr}_{0.1}\text{O}_{1.45}$  were prepared by the Pechini method (14). Nitrates of the required oxides were dissolved in distilled water and citric acid (2 moles per mole perovskite) was added to the solution. The solution was heated to 90°C and ethylene glycol was added while stirring. Further heating resulting in the evaporation of water followed by the emission of  $\text{NO}_x$  fumes and subse-

quent formation of a resin. The resin was charred at 450°C and crushed to yield a fine powder. For  $\text{LaYO}_3$  the powder was sintered at 650°C for 12 hr to form a bixbyite phase of the nominal composition  $\text{LaYO}_3$ . For  $\text{La}_{0.9}\text{Sr}_{0.1}\text{O}_{1.45}$ , the powder was sintered at 800°C for 12 hr. This procedure ensures adequate mixing of cations when the final firing temperature is to be maintained below 800°C.

#### CHARACTERIZATION

##### (i) Identification of Phases

A Diano X-ray diffractometer was used to identify the phases which were present in the prepared oxides. A copper  $K_\alpha$  radiation passed through a nickel filter was used as the source beam. All the samples used in this experiment were found to consist of the desired single phase within the limitations of the X-ray diffraction technique.

##### (ii) Measurement of Surface Areas

The surface areas of the prepared samples were measured by the single-point BET method using a Quantachrome Monosorb analyzer. Table 1 lists the prepared oxides and the surface area of each oxide.

##### (iii) Catalytic Experiments

Catalysts were tested in a fixed-bed quartz reactor (10 mm O.D) for the methane coupling reaction at a constant temperature of 795°C under steady-state conditions. This was selected to avoid complications arising from homogeneous gas-phase reactions. The apparatus has been described and calibrated elsewhere (16). The feed gas composition was 30%  $\text{CH}_4$ , 5%  $\text{O}_2$ , and 65% He. The total input feed rate was maintained constant at 100  $\text{cm}^3/\text{min}$ . A catalyst weight of 0.4 g was used for the reaction in each case.

##### (iv) Product Analysis

The unreacted feed gases and the products were analyzed by an on-line gas chromatograph using a thermal conductivity de-

TABLE I  
List of Catalysts Examined for the Methane Coupling Reaction

Catalyst	Method of preparation	Sintering temp. (°C)	Structure	Conduction type	Surface area (m <sup>2</sup> /gm)
La <sub>0.8</sub> Sr <sub>0.2</sub> CrO <sub>3-δ</sub>	Solid state	1100	Perovskite	p-Type	0.94
La <sub>0.8</sub> Sr <sub>0.2</sub> MnO <sub>3-δ</sub>	Solid state	1100	Perovskite	p-Type	1.36
La <sub>0.8</sub> Sr <sub>0.2</sub> CoO <sub>3-δ</sub>	Solid state	1100	Perovskite	p-Type	0.20
LaFe <sub>0.8</sub> Nb <sub>0.2</sub> O <sub>3-δ</sub>	Solid state	1100	Perovskite	n-Type	1.01
La <sub>1.8</sub> Sr <sub>0.2</sub> CuO <sub>4-δ</sub>	Solid state	950	K <sub>2</sub> NiF <sub>4</sub>	p-Type	0.80
Nd <sub>2</sub> CuO <sub>4-δ</sub>	From (15)	—	Close to K <sub>2</sub> NiF <sub>4</sub>	n-Type	1.04
Nd <sub>1.8</sub> Ce <sub>0.2</sub> CuO <sub>4-δ</sub>	From (15)	—	Close to K <sub>2</sub> NiF <sub>4</sub>	n-Type	0.28
LaYO <sub>3</sub> -bixbyite	Pechini	650	Bixbyite	Insulator	5.4
LaYO <sub>3</sub> -perovskite	Solid state	1100	Perovskite	Insulator	1.08
La <sub>0.8</sub> Sr <sub>0.2</sub> YO <sub>2.9</sub>	Solid state	1100	Perovskite	Electrolyte	1.1
La <sub>0.9</sub> Sr <sub>0.1</sub> O <sub>1.45</sub>	Pechini	800	Hexagonal	—	1.44
La <sub>2</sub> O <sub>3</sub>	From manufacturer	—	Hexagonal	Insulator	2.4
Y <sub>2</sub> O <sub>3</sub>	From Y(NO <sub>3</sub> ) <sub>3</sub>	800	Bixbyite	Insulator	6.4
SrO	From SrCO <sub>3</sub>	800	NaCl	Insulator	0.8

tector with Carbosphere and Haysep column packings. The outlet stream gases were analyzed for CO, CO<sub>2</sub>, CH<sub>4</sub>, C<sub>2</sub>H<sub>4</sub>, C<sub>2</sub>H<sub>6</sub>, and O<sub>2</sub>. Any H<sub>2</sub>O formed during the reaction was trapped before sampling the products and was not analyzed in this procedure.

The conversion of methane was calculated as the fraction of methane reacted according to

$$X = \left( 1 - \frac{\text{moles CH}_4 \text{ out}}{\text{moles CH}_4 \text{ in}} \right). \quad (1)$$

The yields,  $\Sigma\text{CO}_x$  (sum of CO and CO<sub>2</sub>) and  $\Sigma\text{C}_2$  (sum of C<sub>2</sub>H<sub>4</sub> and C<sub>2</sub>H<sub>6</sub>), were obtained as follows:

$$\Sigma\text{CO}_x = \frac{\text{moles CO} + \text{moles CO}_2}{\text{moles CH}_4 \text{ reacted}}, \quad (2)$$

$$\Sigma\text{C}_2 = \frac{\text{moles C}_2\text{H}_4 + \text{moles C}_2\text{H}_6}{\text{moles CH}_4 \text{ reacted}}. \quad (3)$$

#### RESULTS AND DISCUSSION

Table I lists the oxides used in this study along with their structures, conduction type, and processing parameters. Of all the compounds studied, only LaYO<sub>3</sub> has two crystal structures that can be achieved in a

stable form at room temperature. The bixbyite structure of LaYO<sub>3</sub> as in Y<sub>2</sub>O<sub>3</sub> is similar to the CaF<sub>2</sub> fluorite structure but with 25% of the oxygen sites being vacant. It is unlike the perovskite structure, since there is a large concentration of oxygen vacancies.

The perovskites with high p-type electronic conductivity did not yield any C<sub>2</sub> products. The only electronically conducting perovskites with C<sub>2</sub> yield were La<sub>0.8</sub>Sr<sub>0.2</sub>MnO<sub>3-δ</sub> and LaFe<sub>0.8</sub>Nb<sub>0.2</sub>O<sub>3-δ</sub>, both of which are probably n-type under the conditions of the experiment. From Table I, only those compositions with any significant C<sub>2</sub> yield are presented here. The concentrations of the different gases in the outlet stream and the percentage CH<sub>4</sub> converted are given in Table 2 for such compositions. The two perovskites with high electronic conduction show small C<sub>2</sub> yields and high CO<sub>x</sub> yields, whereas the oxides in the La-Sr-Y-O system all show significant C<sub>2</sub> yields. LaFe<sub>0.8</sub>Nb<sub>0.2</sub>O<sub>3-δ</sub> is believed to be an n-type conductor (8) under the reactor conditions of oxygen partial pressure. While La<sub>0.8</sub>Sr<sub>0.2</sub>MnO<sub>3-δ</sub> is a p-type conductor in air, it is likely to be an n-type conductor under reducing conditions due to the stabil-

TABLE 2  
Composition of Outlet Stream and Calculated Yields from Products

Catalyst	% CH <sub>4</sub> converted	Product (%)				Yield		
		CO	CO <sub>2</sub>	C <sub>2</sub> H <sub>4</sub>	C <sub>2</sub> H <sub>6</sub>	ΣCO <sub>r</sub>	ΣC <sub>2</sub>	C <sub>2</sub> <sup>r</sup> /C <sub>2</sub>
La <sub>0.8</sub> Sr <sub>0.2</sub> MnO <sub>3-δ</sub>	11.9	0.15	3.00	0.10	0.26	80.56	18.35	0.39
LaFe <sub>0.8</sub> Nb <sub>0.2</sub> O <sub>3-δ</sub>	11.4	0.21	2.74	0.08	0.26	80.57	18.88	0.32
LaYO <sub>3</sub> -bixbyite	16.8	0.57	2.27	0.54	0.68	52.99	45.69	0.85
LaYO <sub>3</sub> -perovskite	14.4	0.57	2.60	0.15	0.45	71.22	27.12	0.45
La <sub>0.8</sub> Sr <sub>0.2</sub> YO <sub>2.9</sub>	15.4	0.35	2.44	0.42	0.68	55.50	43.70	0.61
La <sub>0.9</sub> Sr <sub>0.1</sub> O <sub>1.45</sub>	18.3	0.28	2.06	0.75	1.05	38.79	59.59	0.72
La <sub>2</sub> O <sub>3</sub>	17.5	0.36	2.20	0.63	0.84	45.95	52.61	0.75
Y <sub>2</sub> O <sub>3</sub>	17.2	0.78	2.55	0.52	0.73	61.13	35.96	0.87

Note. Feed is 30% CH<sub>4</sub> + 5% O<sub>2</sub> + 65% He. Temperature = 795°C.

ity of Mn<sup>2+</sup> ion. p-type conductors gave no C<sub>2</sub> yields, whereas n-type conductors exhibited small C<sub>2</sub> yields. The superconducting oxides were not found to be active for the coupling reaction.

The equilibrium composition for a feed gas containing 30% CH<sub>4</sub> + 5% O<sub>2</sub> + 65% He was calculated theoretically at 795°C. The results are displayed in Table 3. According to these results, the carbon activity in the solid state is equal to one, therefore there should be carbon deposition in solid form as coke. During our experiments, carbon deposition was occasionally observed on the catalysts. The theoretical equilibrium

oxygen partial pressure for the feed composition calculated from the equilibrium concentrations of CO and CO<sub>2</sub> is  $4 \times 10^{-23}$  atm.

In contrast, the concentrations of CO and CO<sub>2</sub> in the outlet stream show that the oxygen partial pressure is actually much higher than the theoretical equilibrium value. The experimentally observed oxygen partial pressure calculated from the outlet stream composition is a maximum of  $1.37 \times 10^{-16}$  atm for La<sub>0.8</sub>Sr<sub>0.2</sub>MnO<sub>3-δ</sub> and a minimum of  $3.65 \times 10^{-18}$  atm for Y<sub>2</sub>O<sub>3</sub>. In all cases, the observed oxygen pressure is much higher than the theoretical oxygen partial pressure from thermodynamic calculations. The experimental oxygen partial pressure is still, however, quite reducing.

Among the compounds in the La-Y-O, strontium-substituted lanthanum yttrium oxide, La<sub>0.8</sub>Sr<sub>0.2</sub>YO<sub>2.9</sub>, with the perovskite structure is an ionic conductor (17) with an ionic conductivity similar to that of the well known electrolyte, stabilized zirconia. The substitution of strontium for lanthanum in these compounds introduces oxygen vacancies in these materials. For a 20 mole% substitution of strontium for lanthanum, there are 3.33% oxygen vacancies introduced on the oxygen sublattice. Therefore both La<sub>0.8</sub>Sr<sub>0.2</sub>YO<sub>2.9</sub> and La<sub>0.9</sub>Sr<sub>0.1</sub>O<sub>1.45</sub> have a considerable anionic fraction of oxygen vacancies. Sr-substituted lanthanum oxides are also ionic conductors with conductivity

TABLE 3  
Calculated Equilibrium Composition of Gas for 30% CH<sub>4</sub> + 5% O<sub>2</sub> + 65% He Input at 795°C

Species	Mole fraction in product (%)
CO(g)	6.938
CO <sub>2</sub> (g)	0.075
CH <sub>4</sub> (g)	0.902
C <sub>2</sub> H <sub>4</sub> (g)	$1.50 \times 10^{-5}$
C <sub>2</sub> H <sub>6</sub> (g)	$0.59 \times 10^{-5}$
C <sub>3</sub> H <sub>4</sub> (g)	$2.52 \times 10^{-9}$
C <sub>3</sub> H <sub>8</sub> (g)	$0.13 \times 10^{-9}$
H <sub>2</sub> O(g)	0.423
H <sub>2</sub> (g)	42.84
O <sub>2</sub> (g)	$3.24 \times 10^{-21}$
C (cryst. + liq.)	Activity = 1.00

TABLE 4  
Diffusion Coefficients of Selected Oxides

Composition	Temperature (°C)	Chemical diffusion coefficient ( $\text{cm}^2/\text{s}$ )
$\text{La}_{0.9}\text{Sr}_{0.1}\text{CoO}_{3-\delta}$ (21)	850	$6.3 \times 10^{-6}$
$\text{La}_{0.9}\text{Sr}_{0.1}\text{FeO}_{3-\delta}$ (21)	900	$7.41 \times 10^{-6}$
$\text{La}_{0.79}\text{Sr}_{0.2}\text{MnO}_{3-\delta}$ (22)	860	$1.7 \times 10^{-6}$
$\text{La}_{0.8}\text{Sr}_{0.2}\text{YO}_{3-\delta}$	1000	$1 \times 10^{-5}$
$\text{Y}_2\text{O}_3$ (23)	1000	$2.63 \times 10^{-9}$

<sup>a</sup> Estimated from conductivity data of (17).

close to zirconia between 600 and 800°C (18).

The diffusion coefficient of oxygen in the electrolyte,  $\text{La}_{0.8}\text{Sr}_{0.2}\text{YO}_{2.9}$ , is greater than that of the manganite,  $\text{La}_{0.79}\text{Sr}_{0.2}\text{MnO}_{3-\delta}$  (Table 4), which is consistent with the greater  $\text{C}_2$  yield of the  $\text{La}_{0.8}\text{Sr}_{0.2}\text{YO}_{2.9}$  solid solution if mobility of oxygen vacancies were important for coupling reactions. However, the diffusion coefficients of  $\text{Y}_2\text{O}_3$  is considerably lower than that of  $\text{La}_{0.79}\text{Sr}_{0.2}\text{MnO}_{3-\delta}$  and the other perovskites, while the  $\text{C}_2$  yield is found to be much greater for  $\text{Y}_2\text{O}_3$ . The diffusion coefficient of oxygen, then, is not the only important factor for high  $\text{C}_2$  yields.

For unsubstituted  $\text{LaYO}_3$ , it has been reported (19) that the bixbyite phase has a high  $\text{C}_2$  yield whereas the perovskite phase has almost no  $\text{C}_2$  yield for a feed gas containing 13.3%  $\text{CH}_4$  + 6.65%  $\text{O}_2$  + 80% He ( $\text{CH}_4/\text{O}_2 = 2$ ) and also for  $\text{CH}_4/\text{O}_2 = 5$  at 700°C. Our results indicate a significant  $\text{C}_2$  yield for the perovskite as well as bixbyite phases both at 700 and 795°C. The strontium-substituted compound,  $\text{La}_{0.8}\text{Sr}_{0.2}\text{YO}_{2.9}$ , also is a perovskite with significant  $\text{C}_2$  yield, as can be seen from Table 2. Therefore from our results, the basic crystal structure is only one important factor for  $\text{C}_2$  yields, as was suggested by (19).

The substitution of strontium in  $\text{LaYO}_3$  increases the concentration of oxygen vacancies in the electrolyte without making it an electronic conductor. A subsequent increase in  $\text{C}_2$  yield is observed. In the other strontium-substituted perovskites, stron-

tium addition not only increases the oxygen vacancies but also increases their electronic conductivity, and the  $\text{C}_2$  yields are found to be smaller or negligible. Therefore a combination of an insulator or electrolyte combined with increased oxygen vacancy concentration appears to increase the  $\text{C}_2$  yield. The  $\text{La}_{0.8}\text{Sr}_{0.2}\text{YO}_{2.9}$  catalyst also exhibited the highest activity for total oxidation of methane among similar types of perovskite oxides (20).

## CONCLUSIONS

A study of the coupling reactions on perovskite oxides,  $\text{K}_2\text{NiF}_4$ -type superconducting oxides, component oxides, and the solid solution  $\text{La}_{0.9}\text{Sr}_{0.1}\text{O}_{1.45}$  has been performed. Materials with n-type conduction, electrolytes, and insulators resulted in the formation of  $\text{C}_2$  products from  $\text{CH}_4$ . Among these, oxides from the La-Sr-Y-O system gave the highest  $\text{C}_2$  yields. Within this system, oxides with a higher oxygen vacancy concentration,  $\text{La}_{0.8}\text{Sr}_{0.2}\text{YO}_{2.9}$  and  $\text{La}_{0.9}\text{Sr}_{0.1}\text{O}_{1.45}$ , were found to be more active for coupling than the component oxides.

## REFERENCES

1. Carberry, J. J., Rajadurai, S., Alcock, C. B., and Li, B., *Catal. Lett.* **4**, 43 (1990).
2. Rajadurai, S., Carberry, J. J., Li, B., and Alcock, C. B., *J. Catal.* **131**, 582 (1991).
3. Barnard, K. R., Foger, K., Turney, T. W., and Williams, R. D., *J. Catal.* **125**, 265 (1990).
4. Mizusaki, J., *Kikan Kagaku Sosetsu* **3**, 1 (1988).
5. Karim, D. P., and Aldred, A. T., *Phys. Rev. B* **20**(6), 2255 (1979).
6. Koc, R., Anderson, H. U., and Howard, S. A., in "Proc. 1st Intern. Symp. SOFCs" (S. C. Singhal, Ed.), The Electrochem. Soc. Proc., Vol. 89-11, pp. 30-40, 1989.
7. Hammouche, A., Schouler, E. J. L., and Henault, M., *Solid State Ionics* **28**, 1205 (1988).
8. Doshi, R., Ph.D. thesis, University of Notre Dame, 1991.
9. Voorhoeve, R. J. H., in "Advanced Materials for Catalysis" (Garten, Ed.), pp. 128-181, 1976.
10. Arai, H., Yamada, T., Eguchi, K., and Seiyama, T., *Appl. Catal.* **26**, 265 (1986).
11. Prasad, R., Kennedy, L. A., and Ruckenstein, E., *Combust. Sci. Technol.* **22**, 271 (1980).
12. Keller, G. E., and Bhasin, M. M., *J. Catal.* **73**, 9 (1982).

13. Ito, T., and Lunsford, J. H., *Nature* **314**, 721 (1985).
14. Pechini, M. P., U.S. Patent 3,330,697, 1967.
15. Smyth, D. M., Lehigh Univ., submitted for publication.
16. Serrano, C., and Carberry, J. J., *Appl. Catal.* **19**, 119 (1985).
17. Wang, Li, Ph.D. thesis, Univ. of Notre Dame, 1991.
18. Vanbaelinghem, F., Pelloux, A., and Deportes, C., *J. Appl. Electrochem.* **6**, 67 (1976).
19. Rehspringer, J. L., Poix, P., Kaddouri, A., Andriamasi-  
masinoro, D., and Kiennemann, A., *Catal. Lett.* **10**, 111 (1991).
20. Doshi, R., Alcock, C. B., Gunasekaran, N., and Carberry, J. J., *J. Catal.* **140**, 557 (1993).
21. Ishigaki, T., Yamauchi, S., Kishio, K., Mizusaki, J., and Fueki, K., *J. Solid State Chem.* **73**, 179 (1988).
22. Belzner, A., Gur, T. M., and Huggins, R. A., in "Proc. 1st Inter. Symp. on SOFCs (S. C. Singhal, Ed.), The Electrochem. Soc., Vol. 89-11, pp. 214-217, 1989.
23. Berard, M. F., Wirkus, C. D., and Wilder, D. R., *J. Am. Ceram. Soc.* **51**, 643 (1968).